A Cyano-Substituted Organoboron Electron-deficient Building Block for D-A Type Conjugated Polymers

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Electronic Supplementary Information

Abstract The development of donor-acceptor (D-A) type conjugated polymers depends largely on the design of novel A building blocks. Herein, we report a novel A building block based on the cyano-substituted organoboron unit (**SBN-3**). Compared with the most common fluorine-substituted $B \leftarrow N$ unit, **SBN-3** displays a significantly downshifted LUMO energy level because of the strong electron-withdrawing ability of cyano groups. In addition, due to the greater impact of cyano substitution on LUMO than on HOMO, **SBN-3** exhibits a reduced band gap, near-infrared absorption and fluorescence properties. The D-A type conjugated polymers based on the cyano-substituted $B \leftarrow N$ unit with thiophene-based units show narrow optical band gaps of *ca*. 1.3 eV as well as distinctive electronic structures, *i.e.*, delocalized LUMOs and localized HOMOs. This work thus provides not only an effective approach to design strong A units but also a new electron-deficient building block for D-A type conjugated polymers.

Keywords Cyano-substituted; B-N unit; Building block; Narrow band gap; D-A type conjugated polymers

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INTRODUCTION

Conjugated polymers with delocalized π -electrons have attracted considerable attention in the last few decades, because of their distinctive optical, electronic and magnetic properties in the neutral or doped state.^[1-4] In particular, D-A type conjugated polymers play an important role in highperformance organic optoelectronic devices, such as organic field-effect transistors (OFETs), organic solar cells (OSCs), and organic photodetectors, due to their great advantages of handily tunable electronic energy levels, broad absorption spectra, narrow band gaps and high charge carrier mobilities.^[5–9] D-A type conjugated polymers are generally constructed by the alternating copolymerization between the electron-rich (D) building blocks and electron-deficient (A) building blocks. The electron-donating and electronwithdrawing capacities of the D and A building blocks, respectively, have a great impact on the energy levels and band gaps of the D-A polymers.^[10] To now, the common D units are often the electron-rich thienyl and phenyl moieties, such as fluorene,^[11] carbazole,^[12] cyclopentadithiophene,^[13] benzodithiophene^[14,15] and indacenodithiophene.^[16,17] The popular A units are mainly based on the strong electronwithdrawing groups, such as imide,^[18,19] amide,^[20,21] thiadiazole,^[22,23] cyano groups,^[24] and boron-nitrogen coordination bonds (B—N).^[25–28] However, the type and number of A units those have been reported so far are still limited because the introduction of strong electronwithdrawing substituent limits the further modifiability of the backbone. Therefore, the development of synthetic strategies for novel A building blocks is important in the diversification of D-A type conjugated polymers.

Organoboron chemistry provides a powerful toolbox for the design of conjugated polymers. These conjugated polymers based on B←N units have already been applied to a variety of organic optoelectronic devices and achieved excellent device performance, such as OSCs, OFETs and organic thermoelectrics.^[26,27,29-33] The electron-deficient nature of B←N units can be attributed to not only the electronic structure of coordination bonds but also the electron-withdrawing substituents on the boron atoms.^[34–36] In the vast majority of cases, the boron atoms are substituted with fluorine atoms.^[26,27,29,37,38] Additionally, a few B←N units are functionalized with phenyl groups on the boron atoms.[28,31] The electron-withdrawing cyano groups, one of the efficient design strategies for A units, can significantly decrease the LUMO and HOMO energy levels of the resulting building blocks when introduced into π -conjugated skeletons.^[39] Thus, by combining the B—N bonds with cyano groups in a π -conjugated skeleton is expected to obtain a stronger A building

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block.

In this study, we focused on the newly developed N−B←N based building block (SBN-1) with a balanced resonance hybrid between B, N covalent and coordination bonds.^[40] The resonance of two B, N bonds endows SBN-1 with excellent chemical stability and photostability. In order to enhance the electron affinity of SBN-1, the nucleophilic substitution reaction with trimethylsilyl cyanide (TMSCN) was conducted, leading to the cyano-substituted SBN-3 (see Fig. 1a). Density functional theory (DFT) calculations and experimental results have confirmed that the replacement of the fluorine atoms with cyano groups can significantly decrease the LUMO and HOMO energy levels of SBN-3. In particular, the LUMO energy level of SBN-3 is 0.36 eV lower than that of SBN-1 as determined by cyclic voltammetry measurement. As a consequence, **SBN-3** exhibits a LUMO energy level as lower as –3.51 eV. In addition, due to the greater impact of cyano substitution on LUMO than on HOMO, SBN-3 exhibits a significantly reduced band gap, near-infrared absorption and fluorescence properties. Considering the strong electron-deficient nature of SBN-3, we transformed it into the di-bromo monomer and further prepared two D-A polymers via Stille polymerization between the resultant monomer and thiophene-based D units. The two polymers both show small optical band gaps of ca. 1.3 eV. This work thus provides not only an effective approach to design strong A units but also a new electron-deficient building block for D-A type conjugated polymers.



Fig. 1 (a) Schematic diagram of the chemical structures of **SBN-1** and **SBN-3**; (b) The calculated LUMO/HOMO energy levels of **SBN-1** and **SBN-3** at the B3LYP/6-31G(d, p) level.

EXPERIMENTAL

Materials

All solvents and reagents were purchased from commercial sources and used without further purification unless otherwise noted. Anhydrous CH_2Cl_2 was used as received. Anhydrous toluene was dried over sodium and distilled prior to use. Compounds **1** and **2** were prepared following the previously reported methods.^[40]

Synthesis

Synthesis of SBN-1

Compound 1 (75 mg, 0.15 mmol) was dissolved in anhydrous

CH₂Cl₂ (10 mL) under an argon atmosphere and heated to 50 °C, and then BF₃·Et₂O (0.77 mL, 5.78 mmol) and Et₃N (0.41 mL, 2.93 mmol) were added dropwise *via* syringe. The resulting mixture was stirred for 4 h at 50 °C. After cooling to room temperature, the mixture was extracted with CH₂Cl₂ and water. The organic layer was washed with brine for three times and dried over Na₂SO₄. After removing the solvent by evaporation, the residue was purified by silica gel column chromatography using CH₂Cl₂/petroleum ether (1:1, *V*:*V*) as the eluent. **SBN-1** was obtained as a dark green solid in 86% yield (76 mg, 0.12 mmol). ¹H-NMR (500 MHz, C₆D₆, δ , ppm): 8.34 (s, 2H), 7.25 (s, 2H), 6.84 (s, 2H), 5.95 (s, 2H), 1.99 (t, *J*=7.5 Hz, 4H), 1.21 (dt, *J*=14.1, 7.0 Hz, 4H), 1.15–0.95 (m, 12H), 0.90 (t, *J*=7.3 Hz, 6H). Anal. Calcd. for C₃₀H₃₄B₂F₄N₄S₂: C, 58.84; H, 5.60; B, 3.53; F, 12.41; N, 9.15; S, 10.47. Found: C, 58.72; H, 5.76; N, 9.04.

Synthesis of SBN-3

To a solution of **SBN-1** (50 mg, 0.08 mmol) in anhydrous CH₂Cl₂ (10 mL) under an argon atmosphere were added SnCl₄ (9.6 µL, 0.08 mmol) and trimethylsilyl cyanide (TMSCN) (0.22 mL, 1.63 mmol) dropwise via syringe. After stirring at 25 °C for 3 h, the reaction was quenched with water and extracted with CH₂Cl₂ (2 imes 25 mL). The combined organic layers were washed with a saturated solution of NaHCO₃ (2×25 mL), distilled water (2×25 mL) and dried over Na₂SO₄. After removing the solvent by evaporation, the residue was purified by silica gel column chromatography using CH₂Cl₂ as the eluent. SBN-3 was obtained as a green solid in 82% yield (43 mg, 0.07 mmol). ¹H-NMR (500 MHz, CDCl₃, δ, ppm): 7.65 (s, 2H), 7.33 (dd, *J*=6.1, 3.4 Hz, 2H), 7.05 (s, 2H), 6.42 (s, 2H), 2.90 (t, J=7.5 Hz, 4H), 1.80-1.73 (m, 4H), 1.48–1.42 (m, 4H), 1.38–1.32 (m, 8H), 0.92 (t, 6H). ¹³C-NMR (101 MHz, CDCl₃, δ, ppm): 162.97, 140.56, 131.85, 131.41, 125.01, 117.68, 117.30, 107.89, 77.36, 77.04, 76.72, 31.28, 30.68, 28.65, 27.38, 22.43, 14.00. Anal. Calcd. for C₃₄H₃₄B₂N₈S₂: C, 63.76; H, 5.35; B, 3.38; N, 17.50; S, 10.01. Found: C, 63.70; H, 5.39; N, 17.45.

Synthesis of compound 3

Compound **3** was prepared and purified in a procedure similar to that of **SBN-1** as a dark green solid in 81% yield. ¹H-NMR (500 MHz, C_6D_6 , δ , ppm): 8.32 (dd, *J*=6.2, 3.2 Hz, 2H), 7.43 (s, 2H), 6.85 (dd, *J*=6.2, 3.2 Hz, 2H), 3.81 (d, *J*=7.4 Hz, 4H), 2.24–2.08 (m, 2H), 1.58–1.29 (m, 64H), 0.95–0.91 (m, 12H). ¹³C-NMR (126 MHz, C_6D_6 , δ , ppm): 158.93, 137.11, 136.35, 133.72, 132.15, 125.50, 120.28, 108.17, 103.13, 79.39, 38.62, 32.35, 31.73, 31.72, 30.66, 30.64, 30.25, 30.22, 30.17, 30.16, 29.89, 29.86, 27.27, 23.14, 23.13, 14.36. Anal. Calcd. for $C_{58}H_{88}B_2Br_2F_4N_4O_2S_2$: C, 58.30; H, 7.42; B, 1.81; Br, 13.37; F, 6.36; N, 4.69; O, 2.68; S, 5.37. Found: C, 58.25; H, 7.39; N, 4.78.

Synthesis of compound 4

Compound **4** was prepared and purified in a procedure similar to that of **SBN-3** as a green solid in 80% yield. ¹H-NMR (500 MHz, C_6D_6 , δ , ppm): 7.89 (dd, *J*=6.1, 3.4 Hz, 2H), 7.43 (s, 2H), 6.74 (dd, *J*=6.2, 3.4 Hz, 2H), 3.79 (d, *J*=7.4 Hz, 4H), 2.07 (s, 2H), 1.44–1.29 (m, 64H), 0.94 (d, *J*=7.0 Hz, 12H). ¹³C-NMR (126 MHz, C_6D_6 , δ , ppm): 159.60, 137.58, 135.41, 134.35, 131.60, 128.44, 128.25, 128.06, 126.03, 118.81, 112.39, 104.69, 79.88, 38.97, 32.55, 31.81, 31.79, 30.85, 30.82, 30.44, 30.42, 30.38, 30.35, 30.07. Anal. Calcd. for $C_{62}H_{88}B_2Br_2N_8O_2S_2$: C, 60.89; H, 7.25; B, 1.77; Br, 13.07; N, 9.16; O, 2.62; S, 5.24. Found: C, 60.75; H, 7.29; N, 9.12.

To a 5 mL microwave vial, compound **4** (57.8 mg, 0.047 mmol), 2,5-bis(trimethylstannyl)thiophene (19.23 mg, 0.047 mmol), Pd₂(dba)₃ (1.3 mg, 0.001 mmol) and P(o-Tol)₃ (3.5 mg, 0.011 mmol) were added. The vial was transferred into a nitrogen filled glovebox and anhydrous toluene (1 mL) was added. Then, the vial was sealed, subjected to a microwave reactor and heated at 120 °C for 2 h. After cooling to room temperature, the reaction mixture was poured into methanol. The resulting precipitate was collected and dried under vacuum to afford **PBN-T** as a brick-red solid in 64% yield (35.2 mg). M_n =3.5 kDa, M_w =6.5 kDa, PDI=1.8. Anal. Calcd for C₆₆H₉₂B₂N₈O₂S₃: C, 69.09; H, 8.08; B, 1.88; N, 9.77; O, 2.79; S, 8.38. Found: C, 69.25; H, 8.12; N, 9.65.

Synthesis of PBN-2T

PBN-2T was prepared from compound **4** (52.0 mg, 0.043 mmol) and 5,5'-bis(trimethylstannyl)-2,2'-bithiophene (20.9 mg, 0.043 mmol) in a procedure similar to that of **PBN-T**. The polymer was obtained as a dark red solid in 96% yield (50.6 mg). M_n =10.3 kDa, M_w =18.0 kDa, PDI=1.7. Anal. Calcd. for C₇₀H₉₄B₂N₈O₂S₄: C, 68.39; H, 7.71; B, 1.76; N, 9.11; O, 2.60; S, 10.43. Found: C, 68.23; H, 7.79; N, 9.23.

RESULTS AND DISCUSSION

Synthesis and Characterization of SBN-3

The chemical structures and synthetic routes of **SBN-3** were depicted in Scheme 1. Starting material **1** was prepared according to literature methods.^[40] Direct borylation of **1** with BF₃·Et₂O and Et₃N afforded **SBN-1**, which might be regarded as a control compound. Subsequently, **SBN-3** was synthesized by a nucleophilic substitution reaction of **SBN-1** with TMSCN in the presence of SnCl₄.^[41,42]

The LUMO and HOMO energy levels of the model compounds of **SBN-1** and **SBN-3** were estimated by DFT calculations at the B3LYP/6-31G(d, p) level. As shown in Fig. 1(b), the replacement of fluorine atoms with cyano groups can decrease the LUMO and HOMO energy levels by 0.79 eV and 0.63 eV, respectively. Thus, the corresponding bandgap is reduced by 0.16 eV. The lower energy levels can be ascribed to the stronger electron-withdrawing character of cyano groups than fluorine atoms. The optimized structures of SBN-3 and **SBN-1** are shown in Fig. S1 (in the electronic supplementary information, ESI). The replacement of fluorine atoms with cyano groups endows SBN-3 with a more planar geometry. A distinguishing characteristic of this unit is the balanced resonance hybrid of the B−N and B←N bond. The optimized geometry of SBN-3 shows approximately equal bond lengths of B-N and B-N, which are close to *ca*. 1.56 Å. This suggests that the cyano groups substitution has a negligible effect on the resonance hybrid of the B-N and $B \leftarrow N$. Interestingly, the cyano-substituted SBN-3 has a large molecular dipole of 12.4 Debye, which is substantially higher compared to the 6.8 Debye of the fluorine-substituted SBN-1. This large dipole is also extremely rare in organic compounds. Furthermore, SBN-3 possesses the similar LUMO/HOMO orbital distribution characteristics as SBN-1, its HOMO and LUMO are well delocalized in the whole conjugated skeleton (Fig. S2 in ESI).^[40] This indicates that the incorporation of cyano groups maintained the excellent electron delocalization characteristic of this B←N unit. The well electron delocalization in SBN-3 is also verified by the nuclear independent chemical shift (NICS) values of the aromatic rings in SBN-3 and SBN-1 (Fig. S3 in ESI). The NICS(1)_{ZZ} values of all the aromatic rings in SBN-3 are nearly equal to those of the aromatic rings in SBN-1, suggesting the two molecules have similar electron delocalization properties.

Fig. 2(a) shows the cyclic voltammetry (CV) curves of **SBN-1** and **SBN-3** in THF solutions. **SBN-1** displays a quasi-reversible reduction wave and two oxidation waves, while after the



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Fig. 2 (a) Cyclic voltammograms of **SBN-1** and **SBN-3**, Fc=ferrocene; (b) Normalized UV/Vis absorption and (c) photoluminescence spectra of **SBN-1** and **SBN-3** in dilute toluene solution $(1 \times 10^{-5} \text{ mol/L})$.

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Compound	λ_{abs}^{a} (nm)	Eg ^{opt a} (eV)	$\lambda_{ m em}$ a (nm)	E _{LUMO} ^b (eV)	E _{HOMO} ^b (eV)	$E_{g}^{\text{elec b}}$ (eV)		
SBN-1 ^[40]	432/614/671	1.78	687	-3.15	-4.80	1.65		
SBN-3	452/676/744	1.58	770	-3.51	-5.01	1.50		
^a Measured in dilute toluene solution; ^b $E_{HOMO}/E_{LUMO} = -(4.80 + E_{onset}^{ox}/E_{onset}) eV.$								

Table 1 Photophysical and electrochemical properties of SBN-1 and SBN-3.

cyano substitution, SBN-3 displays two quasi-reversible reduction waves and two oxidation waves. The reported data of SBN-1 are directly summarized in Table 1. The onset reduction potentials versus Fc/Fc⁺ of SBN-3 are -1.65 and -1.29 V, respectively. The onset oxidation potentials versus Fc/Fc⁺ of SBN-3 are 0.00 and 0.21 V, respectively. According to the onset potentials of the reduction and oxidation, the LUMO and HOMO energy levels of **SBN-3** are estimated to be -3.51 and -5.01 eV, respectively. In comparison to those of SBN-1, the LUMO and HOMO energy levels of SBN-3 are lowered by 0.36 and 0.21 eV, respectively. The downshifted LUMO and HOMO energy levels are due to the stronger electron-withdrawing effect of the cyano groups. Moreover, the cyano groups exhibit a more significant influence on LUMO than HOMO, leading to a smaller electrochemical band gap of 1.50 eV. These results are in accordance with DFT calculation results.

Fig. 2(b) shows the absorption spectra of SBN-1 and SBN-3 in dilute toluene solution, and the corresponding data are collected in Table 1. Both of them display multiple absorption bands in the range of 300 nm to 800 nm. The absorption peaks at long-wavelength region of SBN-1 and SBN-3 are associated with the HOMO \rightarrow LUMO transition according to the TD-DFT calculations (see Table S5 in ESI). Compared with those of **SBN-1**, the absorption peaks of **SBN-3** significantly red-shifted, which are located at 452 nm/620 nm/676 nm/744 nm. According to the onset values of their absorption in solutions, the optical bandgaps are calculated to be 1.78 eV for SBN-1 and 1.58 eV for SBN-3. The optical bandgap of SBN-3 is narrower than that of **SBN-1** by 0.2 eV, which is consistent with the electrochemical band gaps obtained by CV. Fig. 2(c) shows the fluorescence spectra of the two compounds. SBN-3 exhibits an emission peak at 770 nm, which is remarkably redshifted than the emission peak of SBN-1 at 687 nm. These

results indicate that the replacement of the fluorine atoms with cyano groups decreased the band gap of **SBN-3** and realized the absorption and emission in the NIR region.

Synthesis and Characterization of Polymers

Based on the strong electron-deficient property of cyanosubstituted SBN-3 building block, we selected thiophene and dithiophene as the D units to prepare D-A type conjugated polymers. To construct D-A polymers based on SBN-3, it is essential to incorporate reactive groups into its skeleton to obtain the monomer. Scheme 2 represents the chemical structure and synthetic route of the key di-bromo monomer 4. We succeeded to attach bromines to the 5-position of the thiazole. To ensure the solubility of 4 and the resulting polymers, two long branched alkoxy chains were introduced to the central phenyl unit in 4. Subsequently, we prepared two polymers, **PBN-T** and **PBN-2T**, via Stille polymerization of **4** and ditrimethylstannnyl thiophene or ditrimethylstannnyl 2,2'bithiophene. According to the gel permeation chromatography (GPC), the number average molecular weight (M_n) and polydispersity index (PDI) are 3.5 kDa and 1.84 for PBN-T and 10.4 kDa and 1.74 for **PBN-2T**.

It was found that the two polymers can be easily dissolved in common organic solvents, such as toluene, chloroform and chlorobenzene. The decomposition temperature (T_d) of **PBN-T** and **PBN-2T** are both above 220 °C, as revealed by thermal gravimetric analysis (TGA).

DFT calculations at the B3LYP/6-31G(d, p) level were conducted to gain insights into the electronic structures of the polymers based on the cyano-substituted $B \leftarrow N$ unit. To simplify the computation, we chose the dimers of **PBN-T** and **PBN-2T** (**BN-T** and **BN-2T**) as model compounds, in which the long branched alkoxy chains are replaced with methoxyl



Scheme 2 The chemical structures and synthetic routes of the di-bromo monomer **4** and the two polymers, **PBN-T** and **PBN-2T**. Reagents and conditions: (i) $BF_3 \cdot Et_2O$, Et_3N , CH_2Cl_2 , 50 °C; (ii) $SnCl_4$, TMSCN, CH_2Cl_2 , 25 °C; (iii) tris(dibenzylideneacetone)dipalladium(0), P(o-tol)_3, toluene, 120 °C, microwave-assisted, 300 W.



Fig. 3 (a) Kohn-Sham LUMO/HOMO of the model compound of **PBN-2T**; (b) Cyclic voltammograms of **PBN-T** and **PBN-2T** as thin film; (c) Normalized UV/Vis absorption of **PBN-T** and **PBN-2T** in dilute toluene solution $(1 \times 10^{-5} \text{ mol/L})$.

Table 2 Photophysical and electrochemical properties of PBN-T and PBN-2T.										
Polymer	$\lambda_{ m abs}$ ^{sol a} (nm)	$\varepsilon^{\mathrm{a,b}}$ (M ⁻¹ ·cm ⁻¹)	$\lambda_{ m abs}^{ m film}$ (nm)	E_{g}^{optc} (eV)	E _{LUMO} ^d (eV)	E _{HOMO} ^d (eV)				
PBN-T	505/757/838	22, 587	505/754/829	1.35	-3.69	-5.11				
PBN-2T	567/768/851	41, 255	562/775/861	1.31	-3.77	-5.23				
^a Moscurod in diluto t	aluana calution ^b Data	/E _ (1 00 E	ox/E red ov							

^a Measured in dilute toluene solution; ^b Determined at the highest peak wavelength; ^c Measured in thin film; ^d $E_{HOMO}/E_{LUMO} = -(4.80 + E_{onset}^{ox}/E_{onset}^{red}) eV$.

groups. **BN-T** and **BN-2T** exhibit similar electronic structures in their LUMO and HOMO distributions, as shown in Fig. 3(a) and Fig. S5 (in ESI). Their LUMOs are delocalized evenly along the backbone, while the HOMOs are mainly localized on the cyano-substituted B—N unit. These characteristics are rare in D-A polymers, which usually show delocalized HOMOs and localized LUMOs.^[15,43,44] The electronic structure of **BN-T** and **BN-2T** is anticipated to significantly impact their opto-electronic properties.

Fig. 3(b) shows the CV curves of **PBN-T** and **PBN-2T**. The two polymers both possess multiple irreversible oxidation waves and reversible reduction waves. The LUMO and HOMO energy levels estimated from the onset reduction and oxidation waves are -3.69 and -5.11 eV for **PBN-T** and -3.77 and -5.23 eV for **PBN-2T**. Thus, the electrochemical band gaps of **PBN-T** and **PBN-2T** are calculated as 1.42 and 1.46 eV. The LUMO energy levels of both polymers are much lower than the HOMO energy levels, compared to the compound **SBN-3**. This difference is due to the delocalized LUMOs and localized HOMOs of both polymers, which has been demonstrated by B \leftarrow N based polymers with similar electronic structures.^[40,45]

Fig. 3(c) shows the absorption spectra of PBN-T and PBN-2T in dilute toluene solution, the corresponding data are summarized in Table 2. The two polymers display similar spectra shapes as the small molecule SBN-3. The maximum absorption peak of PBN-T is located at 507 nm and the longest-wavelength absorption peak is at 836 nm. While, PBN-2T shows an absorption maximum at 567 nm with the longest-wavelength absorption peak at 851 nm. The absorption at 700–900 nm for PBN-T is stronger than that of PBN-2T owing to the higher overlap extent between the LUMO and HOMO distributions for PBN-T (Fig. 3a and Fig. S5 in ESI).^[46] The optical band gaps are 1.35 eV for **PBN-T** and 1.31 eV for PBN-2T as calculated from the absorption onsets of the thin films (see Fig. S6 in ESI). Compared with the small molecule SBN-3, the two polymers exhibit smaller optical band gaps owing to the extended π -conjugation in the polymers. These results imply that the use of the newly developed cyano-substituted B—N unit as the A building block is an effective way to construct D-A type conjugated polymers with narrow band gaps.

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CONCLUSIONS

In summary, we developed a cyano-substituted organoboron electron-deficient building block for D-A type conjugated polymers. The replacement of fluorine atoms with cyano groups at the boron centers endows SBN-3 with a lower LUMO energy level of -3.51 eV owing to the strong electron-withdrawing ability of cyano groups. Furthermore, SBN-3 exhibits a considerably decreased band gap, due to the more substantial influence of cyano substitution on LUMO than on HOMO, leading to NIR absorption and fluorescence characteristics. By using the cyano-substituted B←N unit as the A building block, two D-A conjugated polymers have been further prepared. The resulting polymers exhibit narrow optical band gaps of ca. 1.3 eV as well as distinctive electronic structures, i.e., delocalized LUMOs and localized HOMOs. This work thus provides not only an effective approach to design strong acceptor units but also a new electron-deficient building block for D-A type conjugated polymers.

NOTES

The authors declare no competing financial interest.

Electronic Supplementary Information

Electronic supplementary information (ESI) is available free of charge in the online version of this article at http://doi.org/10.1007/s10118-023-2940-4.

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